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博士論文



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分子與金屬介面的組態與磁性之第一原理研究

First-principles study on the configurations and magnetic  
properties of molecules and metal interfaces

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## 摘要

這篇論文使用基於泛含密度理論的第一原理計算分子與金屬材料介面或金屬與金屬介面間的物理性質，特別著重在不同磁性組態的行為，旨在了解一些特殊物理性質背後的物理原理。文章研究的系統有三種：1. 由兩層磁性金屬夾著一層非磁性金屬構成的巨磁阻系統；2. 將銅沉積在鉻表面形成的合金；3. 具有手性的勝肽（peptide）分子在帶磁性的鈷鍍金表面的吸附力研究。

早期的巨磁阻系統研究發現，該系統兩側金屬的磁性方向可能不是整齊的相互平行或相互反平行，而是帶有部分垂直分量。為此，我們計算了不同厚度的中間非磁性金屬層的層間交換耦合（interlayer exchange coupling），以了解兩側磁性金屬磁矩方向選擇的特性。此外，在兩側磁矩相互垂直的系統中，我們發現了成對出現的特殊能帶，這些能帶在實空間的磁矩沿垂直介面方向呈現順時針方向和逆時針方向旋轉。

實驗發現，銅在鉻表面沉積後形成的二銅一鉻（ $\text{BiCu}_2$ ）合金，在施加特定偏壓時，該合金的電導會突然大幅上升。我們嘗試通過第一原理計算此系統的能帶結構及態密度，以材料的物理性質從理論上理解電導大幅上升的原因。

實驗測量了具有手性的勝肽分子在磁性表面的吸附力，發現不同磁性方向（進表面或出表面）之間存在相當強度的差距。利用第一原理計算，我們計算此系統的能量，試圖探討吸附力差距的來源及其物理機制。

關鍵字：第一原理、磁矩、層間交換耦合、非共線排列、巨磁阻系統、拉什巴效應、手性誘導自旋選擇性





# Abstract

This dissertation employed first-principles calculations to study molecule-metal and metal-metal interfaces, with a particular focus on their configurations and magnetic properties. The objective was to elucidate the physical mechanisms underlying unusual phenomena. Three systems are studied in this research: 1. a giant magnetoresistive system consisting of two layers of magnetic metal sandwiching a non-magnetic metal spacer; 2. an alloy formed by depositing Cu on a Bi surface; and 3. the adsorption of a chiral peptide molecule on Au/Co surfaces with magnetic moment.

Initial studies of the giant magnetoresistive system indicated that the direction of metallic magnetism in both side layers may not be perfectly parallel or antiparallel, but may have a partially perpendicular component. To address this, we calculate the interlayer exchange coupling (IEC) of the system with different thicknesses of the non-magnetic spacer to improve our understanding of its properties concerning the orientations of the magnetic moments on both sides. Furthermore, in configurations where the magnetic mo-

ments are perpendicular to each other, we observed a pair of stripes of magnetic moments rotating clockwise and counterclockwise in real space along a direction perpendicular to the interface.



The deposition of Cu on the Bi surface results in the formation of the  $\text{BiCu}_2$  alloy, which exhibits a sudden and significant increase in conductivity at a specific bias voltage. To comprehend the underlying physical mechanism responsible for this dramatic increase in conductivity, we have calculated the band structure and density of states of this system from first principles.

The adsorption of chiral molecules on magnetic surfaces was measured and the results showed significant differences in magnetic orientation attraction (in-plane vs. out-of-plane). To understand the physics behind this difference, we performed ab initio calculations to determine the energy of the system.

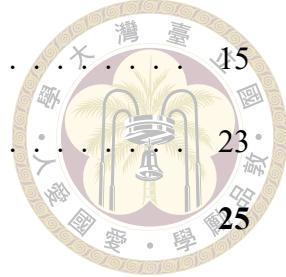
**Keywords:** first-principles, magnetic moment, interlayer exchange coupling, noncollinear alignment, metallic trilayer, Rashiba effect, Chirality-induced spin selectivity (CISS)



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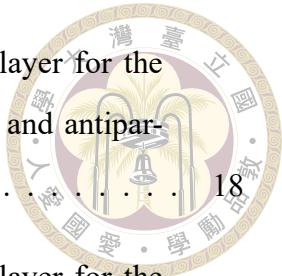


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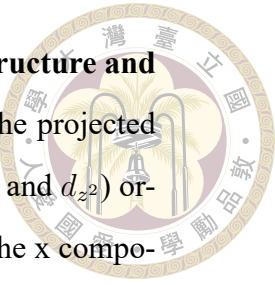


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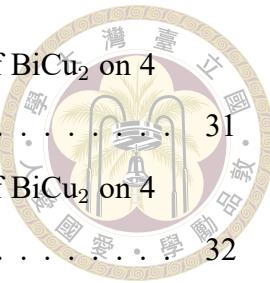


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# Chapter 1 Introduction

As technology advances, the fabrication of nanoscale devices is becoming increasingly feasible. The interfacial properties and quantum effects of the nanoscale devices are critical in determining their physical properties. First-principles calculations can be effectively applied at this scale range. By analyzing the band structure and density of states, the electrical and magnetic properties of the device can be fully understood. In this dissertation, three systems with magnetic moments and interfaces are studied, and the experimentally observed phenomena are elucidated using first-principles calculations.

The first-principles calculations discussed in this dissertation were based on density functional theory (DFT). Chapter 2 provides a brief overview of DFT. The calculations presented in this dissertation were performed using the Vienna Ab initio Simulation Package (VASP). [1–4]

Chapter 3 considers a giant magnetoresistive system consisting of two magnetic metal layers sandwiching a non-magnetic metal spacer. Recent experiments on Co/Ir/Co and Co/Ru/Co trilayers have shown that manipulating the width or properties of the spacer layer can induce noncollinear and even perpendicular magnetic alignment between the Co layers. Using first-principles methods to investigate the interlayer exchange coupling (IEC) in the Fe/Ag/Fe trilayer, we find that the magnetic moments of the two Fe layers tend

to align perpendicularly for a given width of the Ag layer. Our results are in agreement with the analytical calculations performed by a collaborator. [5] We have also found a relationship between the IEC and the magnetic orientation of the system. In particular, unique bands emerge in pairs, and the magnetic moments within these bands show a spiral rotation across the trilayer. These results demonstrate that nonlinear IEC provides a certain degree of freedom in controlling magnetic configuration, thereby increasing the load carrying capacity of spintronics technology.

Cu deposition on the Bi surface results in of the alloy, BiCu<sub>2</sub>, which is discussed in Chapter 4. Under certain voltage conditions, the conductivity of this system increases significantly. We employed first-principles methods to calculate the band structure and density of states for 2 to 8 layers of Bi. However, the projected band structures do not agree well with the previous Korringa-Kohn-Rostoker (KKR) calculations. [6] Nevertheless, I have gained considerable knowledge in this area.

The adsorption forces of chiral peptides on an Au/Co surface were measured, revealing significant differences in the direction of the surface magnetic moment. [7] In Chapter 5, the total energy of chiral peptides on the Au/Co surface is investigated. However, the results do not correlate well with the experimental data, and the reasons for this discrepancy are discussed.



# Chapter 2 Method

## 2.1 Density Functional Theory

In solid state physics, the systems of interest typically consist of many atoms within a repeating unit cell. Consequently, the challenge is to solve the time-independent many-body Schrödinger equation. [8]

$$\hat{H}\psi(\{\vec{r}_i\}, \{\vec{R}_a\}) = E\psi(\{\vec{r}_i\}, \{\vec{R}_a\}) \quad (2.1)$$

$$\begin{aligned} \hat{H} = & \sum_{a=1}^M \frac{-\hbar^2}{2M_a} \nabla_a^2 + \sum_{a=1}^{M-1} \sum_{b=a+1}^M \frac{Z_a Z_b e^2}{R_{ab}} & (\hat{H}_{nucleus}) \\ & + \sum_{i=1}^N \frac{-\hbar^2}{2M_e} \nabla_i^2 + \sum_{i=1}^{N-1} \sum_{j=i+1}^N \frac{e^2}{r_{ij}} & (\hat{H}_{electron}) \\ & - \sum_{a=1}^M \sum_{i=1}^N \frac{Z_a e^2}{r_{ai}} & (\hat{H}_{electron-nucleus}) \\ & + V(\{r_i\}, \{R_a\}) \end{aligned} \quad (2.2)$$

where,  $\vec{r}_i$  is the position vector for the  $i$ -th electron,  $\vec{R}_a$  is the position vector for the  $a$ -th nucleus,  $r_{ai}$  is the distance between the  $a$ -th nucleus and the  $i$ -th electron,  $Z_a$  is the charge of the  $a$ -th nucleus, and  $M_a$  is the mass of the  $a$ -th nucleus,  $M_e$  is the mass of the electron. However, there is no general analytical solution to the Schrödinger equation for three or more bodies. We can use the Born-Oppenheimer approximation, where the nuclei are



relatively fixed, to rewrite the equation

$$\hat{H}\psi(\{\vec{r}_i\}) = \left[ \sum_{i=1}^N \frac{-\hbar^2}{2M_e} \nabla_i^2 + \sum_{i=1}^{N-1} \sum_{j=i+1}^N \frac{e^2}{r_{ij}} + V'(\{r_i\}) \right] \psi(\{\vec{r}_i\}) \quad (2.3)$$

, but only the electric part of the equation is still difficult to solve. In 1965, Walter Kohn and Lu Jeu Sham developed a theorem that simplifies the many-particle equation to a collection of one-particle equations. [9, 10] The first Kohn-Sham theorem states that the ground state energy of the Schrödinger equation is a unique functional of the electron density, that is

$$E_0 = E[\rho_0(\vec{r})] \quad (2.4)$$

$$\rho(\vec{r}) = \sum_{i=1}^N |\psi(\{\vec{r}_i\})|^2 \quad (2.5)$$

where,  $\rho_0(\vec{r})$  is the ground state electron density.

The second Kohn-Sham theorem is like the variational principle, which states that

$$E[\rho(\vec{r})] \geq E_0. \quad (2.6)$$

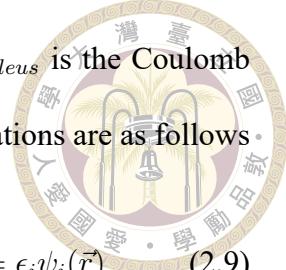
The qual sign holds when

$$\rho(\vec{r}) = \rho_0(\vec{r}). \quad (2.7)$$

The energy functional can be written as:

$$\begin{aligned} E[\rho(\vec{r})] &= T_{electron}[\rho(\vec{r})] + \frac{e^2}{2} \int \frac{\rho(\vec{r})\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d^3r d^3r' + \int V_{electron-nucleus}(\vec{r})\rho(\vec{r})d^3r \\ &+ E_{nucleus} + E_{XC}[\rho(\vec{r})] \end{aligned} \quad (2.8)$$

where  $T_{electron}$  is the kinetic energy of the electron and  $V_{electron-nucleus}$  is the Coulomb potential between the electron and the nucleus. The Kohn-Sham equations are as follows.



$$[T_{electron} + V_H(\vec{r}) + V_{electron-nucleus}(\vec{r}) + V_{XC}(\vec{r})]\psi_i(\vec{r}) = \epsilon_i\psi_i(\vec{r}) \quad (2.9)$$

$$V_H(\vec{r}) = e^2 \int \frac{\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d^3r' \quad (\text{Hartree potential}) \quad (2.10)$$

$$V_{XC}(\vec{r}) = \frac{\delta E_{XC}(\vec{r})}{\delta \rho(\vec{r})} \quad (2.11)$$

## 2.2 Exchange-Correlation Functional

The kinetic energies of the electrons, the electron-electron Coulomb potential, and the electron-nuclear Coulomb potential are well defined in the Kohn-Sham equations. The exchange-correlation energy,  $E_{XC}$ , is the only term that is difficult to understand because it includes all the remaining electronic energy components except the non-interacting kinetic and Coulomb potential terms. Although the precise form of the  $E_{XC}$ , remains unknown, accurate approximations of  $E_{XC}$  can provide reliable total energy values and other physical properties.

The simplest way to approximate the exchange correlation energy is to assume that the local electron density behaves like a uniform electron gas. Then, we have

$$E_{XC}^{LDA} = \int \epsilon_{XC}(\rho(\vec{r}))\rho(\vec{r})d^3\vec{r} \quad (2.12)$$

where the  $\epsilon_{XC}$  is the exchange-correlation energy density, which is only a function of the electronic density. This is known as the local density approximation (LDA). In practical applications, LDA agrees well with many experimental results.

Another widely used approximation is the general gradient approximation (GGA), which considers the exchange-correlation energy density as a function of both the electronic density and its gradient. The  $E_{XC}$  can be written as



$$E_{XC}^{GGA} = \int \epsilon_{XC}(\rho(\vec{r}), \nabla \rho(\vec{r})) \rho(\vec{r}) d^3 \vec{r} \quad (2.13)$$

The GGA is usually better than the LAD for the transition metal with higher angular momentum electrons.

## 2.3 The Self-Consistent Cycle

The Kohn-Sham equations were solved using an iterative method. The self-consistent cycle in VASP is illustrated in Figure 2.1. First, the trial charge density and trial wavevector were generated. The Hamiltonian was then obtained from the previous input. Iterative refinement was then applied to the Kohn-Sham equations to solve for the new wavevectors. With the updated wavevectors, the total energy was calculated. Following this, the initial charge density was mixed with the charge density generated by the new wavevectors to calculate the new charge density. The final step in the electronic self-consistent cycle is to compare the energy of the current iteration with that of the previous loop. If the difference exceeds the required accuracy, a new loop is initiated; otherwise, the loop is closed, and the forces are calculated to update the positions of the ions.

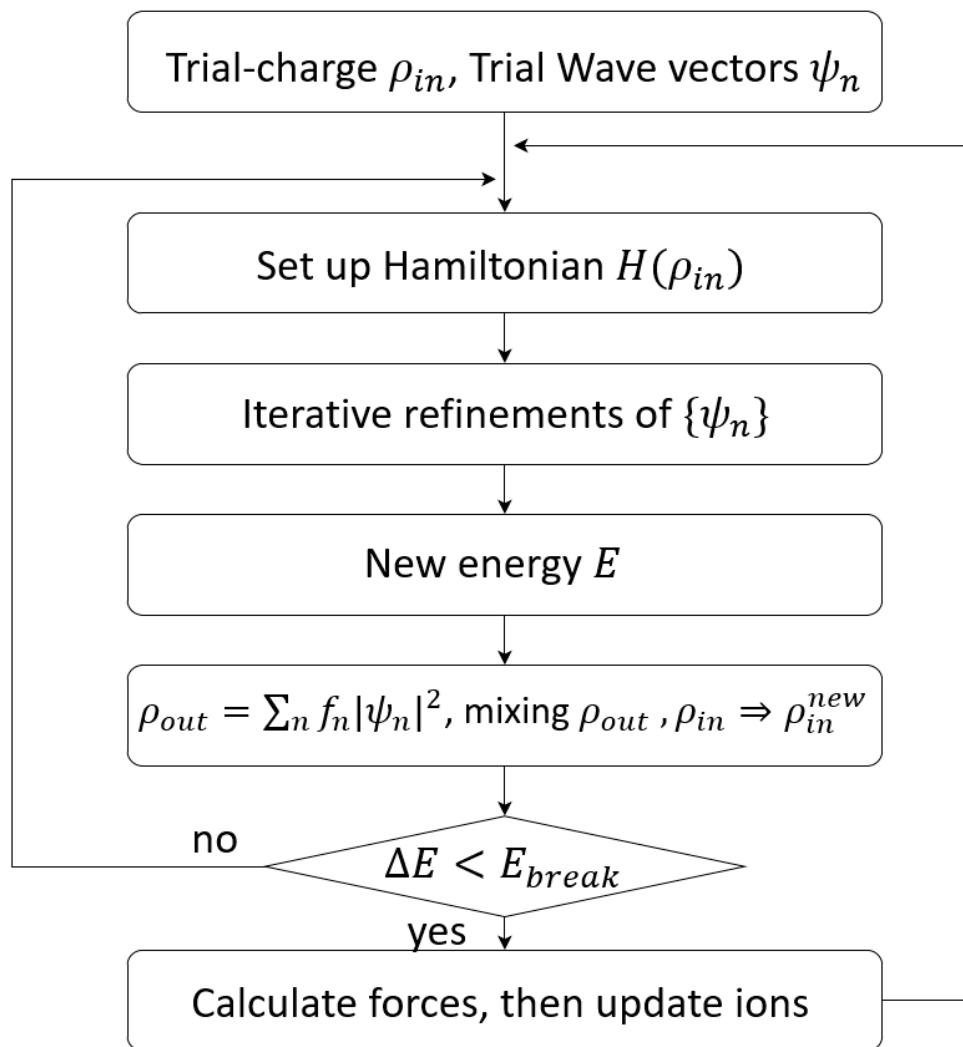


Figure 2.1: The flowchart of the ground state calculation.





# Chapter 3 Driving Noncollinear Interlayer Exchange Coupling Intrinsically in Magnetic Trilayers

## 3.1 Introduction

Giant magnetoresistance (GMR) systems, consisting of spacer layers sandwiched between ferromagnetic layers, are widely used to fabricate a variety of spintronic devices, including spin-torque oscillators, [11, 12] magnetic random access memories,[13–15] magnetic sensors [16, 17], and racetrack memories. [18–20] Most of these devices consist of two or more magnetic layers separated by spacer layers, and interlayer exchange coupling plays an important role in the development of spintronic devices due to its fundamental properties. [21–26]

Recent experiments on interlayer exchange couplings in metallic multilayer structures have shown that the bilinear coupling strength ( $J_1$ ) and the biquadratic coupling strength ( $J_2$ ) have been accurately measured in the Co/IrFe/Co structure. [27, 28] Furthermore, the noncollinear coupling has been observed in the Co/RuFe/Co structure. Since  $J_1$

and  $J_2$  vary with the thickness of the nonmagnetic intermediate layer, not only antiferromagnetic and ferromagnetic configurations of spin configurations between ferromagnetic layers can be controlled, but also non-collinear configurations can be achieved. [27] The potential for non-trivial spin alignment through noncollinear coupling holds great promise for increasing the speed and efficiency of spintronic devices. [27, 28]

Although noncollinear alignment has been observed in GMR systems, including Co/Cu/Co [29] and Fe/Cu/Fe, [30, 31] for more than 30 years. Additionally, it has been found that interface defects and film quality have hampered experimental studies of interlayer exchange coupling. [32–34] In this study, we show that unique noncollinear magnetic orientations can exist in ultrathin magnetic trilayers, consistent with recent experimental observations.  $J_1$  was found to oscillate with spacer thickness, which could be explained by the Ruderman-Kittel-Kasuya-Yosida (RKKY) theory or the quantum well states model. However, the mechanism causing the non-collinear coupling remains poorly understood. The RKKY model is based on a model describing indirect exchange coupling of magnetic impurities within metals, [35–37] while the quantum well model suggests that spin-dependent confinement of electronic states within the spacer layer is responsible for the effective magnetic coupling between magnetic impurities. layer. [23, 38–40]

We propose that the noncollinear couplings term due to IEC,  $J_1$  and  $J_2$ , are demonstrated employing first-principles techniques to a Fe/Ag/Fe trilayer metallic structure. By varying  $J_1$  and  $J_2$  with the thickness of the metallic layer, we provide a theoretical explanation consistent with recent experimental results. We also show the relationship between the ICE and the thickness of the spacer layer. We then studied quantum well states (QWS) in noncollinear systems and found that the spin of the Ag quantum well must rotate in a clockwise or counterclockwise direction perpendicular to the Fe/Ag interface. This ob-

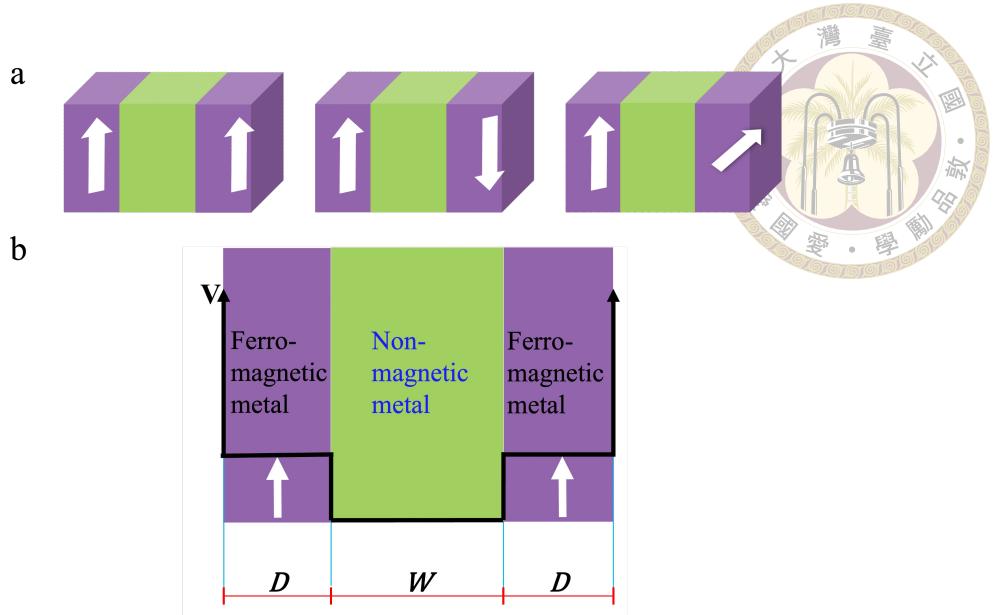


Figure 3.1: **a, Schematic illustration of the magnetic arrangement between ferromagnetic layers.** Parallel, antiparallel and perpendicular arrangements between the side ferromagnetic layers. **b, Schematic diagram of a three-layer quantum well.**

servation shows that the magnetic structure is not collinear and allows spatial rotation of carrier spins in the system, which creates important applications in the development of spintronics.

## 3.2 Quantum-well model

To study the GMR system in the quantum well model, we studied a three-layer metal system consisting of a nonmagnetic silver interlayer of width  $W$  and an iron side layer of width  $D$ , as shown in Figure 3.1. Magnetic coupling is associated with energy changes in different magnetic configurations corresponding to parallel, antiparallel and perpendicular arrangements of the lateral layer. The total energy per unit area of the GMR system is written as: [41]

$$E = \frac{E_F k_F^2}{4\pi^2} \sum_{\sigma=\uparrow,\downarrow} \sum_{n=1}^{\infty} \int d^2 k_{\parallel} \frac{\frac{E(\sigma,n)}{E_F} + \frac{k_{\parallel}^2}{k_F^2} - 1}{\exp[(\frac{E(\sigma,n)}{E_F} + \frac{k_{\parallel}^2}{k_F^2} - 1) \frac{E_F}{k_B T}] + 1}, \quad (3.1)$$

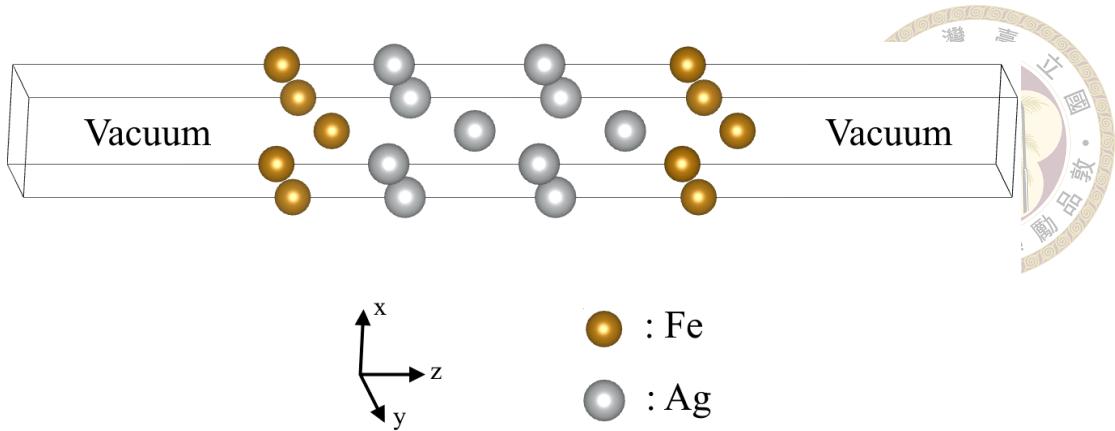


Figure 3.2: Unit cell of the  $\text{Fe}_2/\text{Ag}_4/\text{Fe}_2$  trilayer.

where  $\sigma$  is the spin of the electron (up or down),  $E(\sigma, n)$  is the energy of the  $n$ -th band, ,  $k_{\parallel}$  is the transverse wave vector, and  $T$  is the temperature.

From a phenomenological point of view, the total energy of the system per unit area is given by

$$E = -J_1(\vec{m}_1 \cdot \vec{m}_2) - J_2(\vec{m}_1 \cdot \vec{m}_2)^2, \quad (3.2)$$

where  $J_1$  and  $J_2$  are the IECs, and  $\vec{m}_i$  is normalized magnetic moment of the  $i$ -th ferromagnetic layer. Using Eq. 3.2 to analyse the energy of the system over the three magnetic configurations, the bilinear and biquadratic IECs are as follows

$$\begin{aligned} J_1 &= -\frac{E_{AP} - E_{PA}}{2} \\ J_2 &= -\frac{E_{PA} + E_{AP}}{2} + E_N, \end{aligned} \quad (3.3)$$

In this context,  $E_{AP}$ ,  $E_{PA}$ , and  $E_N$  correspond to the energies per unit area of the antiparallel, parallel, and perpendicular configurations, respectively.



### 3.3 Unit Cell and Calculation Parameters

The lattice structure of bulk Fe is body-centred cubic, whereas that of Ag is face-centred cubic. Ag(001) acts as a spacer between Fe(001) layers rotated at an angle of 45 degrees in the xy plane. The trilayers were separated by a vacuum of at least 18 Å. The  $\text{Fe}_2/\text{Ag}_4/\text{Fe}_2$  trilayer was visualised using VESTA (see Figure 3.2). [42] The layered structure of  $\text{Fe}_2/\text{Ag}_W/\text{Fe}_2$  consists of a BCC-like unit cell consisting of two ferromagnetic monolayers (ML) on both sides and a nonmagnetic Ag layer in the middle. The silver layer thickened from 2 ML to 12 ML. The vacuum required to separate  $\text{Fe}_2/\text{Ag}_W/\text{Fe}_2$  is at least 18 Å. The in-plane lattice constant is 2.89 Å, and which was optimized with Ag-Bulk. The DFT calculations were performed by the Vienna Ab initio Simulation Package (VASP) [2, 43, 44] using the projector augmented wave (PAW) pseudopotentials [44] and the Perdew-Burke-Ernzerhof (PBE) generalized gradient approximation (GGA). The energy convergence criterion is defined as less than  $10^{-3}$  meV. A gamma-centered  $76 \times 76 \times 1$  K-point grid in the Brillouin zone was chosen. The energy limit for the plane wave basis is set to 350 eV. Structurally optimized unit cells were calculated using a force convergence criterion of  $10^{-1}$  meV/Å. Nonlinear collinear calculations for three configurations (parallel, antiparallel and perpendicular orientation, see Figure 3.1) are performed without spin-orbital coupling.

### 3.4 Convergence Tests

To check the convergence of the bilinear interlayer exchange coupling ( $J_1$ ), the collinear calculations of the  $\text{Fe}_2/\text{Ag}_3/\text{Fe}_2$  structure were performed. The energy convergence crite-

Table 3.1: Convergence of ENCUT

ENCUT	$E_{PA}$ (eV/unit cell)	$E_{AP}$ (eV/unit cell)	$J_1$ (meV/unit cell)
270	-37.681299	-37.694752	-13.453
290	-37.710521	-37.724006	-13.485
310	-37.710612	-37.724114	-13.502
330	-37.694477	-37.707983	-13.506
350	-37.677157	-37.690658	-13.501
370	-37.664721	-37.678190	-13.469
400	-37.656335	-37.669804	-13.469
430	-37.653789	-37.667235	-13.446
460	-37.655112	-37.668553	-13.441

Table 3.2: Convergence of NGX(NGY)

NGX(NGY)	$E_{PA}$ (eV/unit cell)	$E_{AP}$ (eV/unit cell)	$J_1$ (meV/unit cell)
18	-37.677157	-37.690658	-13.501
20	-37.677216	-37.690707	-13.491
22	-37.677221	-37.690714	-13.493
24	-37.677241	-37.690737	-13.496
26	-37.677255	-37.690751	-13.496

Table 3.3: Convergence of Kpoints

K	$E_{PA}$ (eV/unit cell)	$E_{AP}$ (eV/unit cell)	$J_1$ (meV/unit cell)
40	-37.653417	-37.667001	-13.584
43	-37.653548	-37.667394	-13.846
46	-37.653506	-37.667407	-13.901
49	-37.653703	-37.667215	-13.512
52	-37.653886	-37.667337	-13.451
55	-37.653556	-37.667301	-13.745
58	-37.653835	-37.667310	-13.475
61	-37.653653	-37.667153	-13.500
64	-37.653911	-37.667387	-13.476

rion is set below  $10^{-4}$  meV.

The parameter ENCUT, the cutoff energy for the plane wave-basis set, is checked for convergence with k-points=76  $\times$  76  $\times$  1. NGX (NGY), is the number of grid points in the fast Fourier transform (FFT) grid along the first (second) lattice vector, is checked with ENCUT=350 eV, also the NGZ is increased to the same ratio. The k-points ( $K \times K \times 1$ ) are checked with NCUT=350 eV. The data of the three parameters mentioned above are shown in Table 3.1-3.3.

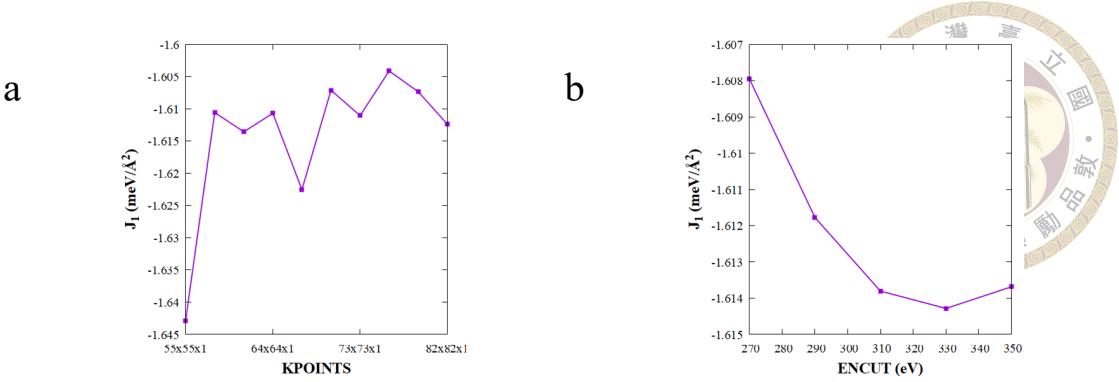


Figure 3.3: a. The  $J_1$  versus the KPOINTS. b. The  $J_1$  versus the ENCUT.

The  $J_1$  calculated by different k-points and ENCUT were plotted in Figs. 3.3a and 3.3b, respectively. The  $J_1$  varied with less than 0.01 meV/Å<sup>2</sup> for ENCUT greater than 270 eV and the k-points greater than 70 × 70 × 1.

### 3.5 Results and Discussion

Table 3.4: The energies of the different magnetic configurations for the Ag layers from 2 to 12.

W	$E_{PA}$ (eV/unit cell)	$E_N$ (eV/unit cell)	$E_{AP}$ (eV/unit cell)
2	-34.956148	-34.953129	-34.955157
3	-37.655260	-37.663186	-37.668944
4	-40.379019	-40.379276	-40.378976
5	-43.123095	-43.121003	-43.119810
6	-45.840901	-45.839443	-45.838395
7	-48.561124	-48.560138	-48.558194
8	-51.275258	-51.275381	-51.276117
9	-53.993685	-53.994611	-53.995850
10	-56.759430	-56.760948	-56.761828
11	-59.487045	-59.486770	-59.486789
12	-62.213760	-62.212677	-62.211832

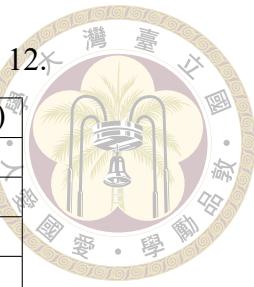
We optimised the Fe<sub>2</sub>/Ag<sub>W</sub>/Fe<sub>2</sub> trilayer structure with the antiparallel alignment of the Fe side layers with collinear spin-polarized calculations. The noncollinear calculation was then performed with the optimised structures. The energies of the different magnetic configurations,  $J_1$ ,  $J_2$ , and  $2|J_2|-|J_1|$ , this determinant will be explained in the following, are listed in Table 3.4 and Table 3.5.

Table 3.5:  $J_1$ ,  $J_2$ , and  $2|J_2|-|J_1|$  for the Ag layers from 2 to 12.

W	$J_1$ (meV/Å <sup>2</sup> )	$J_2$ (meV/Å <sup>2</sup> )	$2 J_2 - J_1 $ (meV/Å <sup>2</sup> )
2	0.118	0.302	0.485
3	-1.636	-0.130	-1.376
4	0.005	-0.033	0.061
5	0.393	0.054	-0.285
6	0.300	0.025	-0.251
7	0.350	-0.057	-0.236
8	-0.103	0.037	-0.029
9	-0.259	0.019	-0.221
10	-0.287	-0.038	-0.210
11	0.031	0.018	0.005
12	0.230	0.014	-0.202

The strengths of  $J_1$  and  $J_2$  in the  $\text{Fe}_2\text{Ag}_W\text{Fe}_2$  trilayer with different spacer widths were shown in Figure 3.4. Figure 3.4a was derived from first-principal calculations, and Figure 3.4b was derived from analytical calculations performed by the collaborator (H.-C Wang). [45] We found that the noncollinear DFT calculation shows that the noncollinear coupling dominates the ground state only in the  $\text{Fe}_2\text{Ag}_4\text{Fe}_2$  structure, i.e. with the specific thickness of the nonmagnetic metal layer. The top part of Figure 3.4a shows ICEs as a function of Ag thickness, derived from Equation 3.3 by cubic spline interpolation. Note that the sign of  $J_1$  directly represents whether parallel or antiparallel magnetic configurations are preferred. As the width of the silver layer increases, the fluctuations of  $J_1$  and  $J_2$  became smaller. The oscillation period of  $J_1$  was almost twice than that of  $J_2$ , and the amplitude of  $J_1$  was approximately seven greater than that of  $J_2$ . With increasing thickness, the amplitude of both  $J_1$  and  $J_2$  decreases. The results of first-principles calculations and analytical calculations are in agreement.

The extreme value of equation 3.2 occurs when  $\vec{m}_1 \cdot \vec{m}_2 = \cos(\theta) = -\frac{J_1}{2J_2}$ , or  $\sin(\theta) = 0$ , where the  $\theta$  is the angle between the magnetic moments of the two iron layers. If the system is not in trivial magnetic alignment, then we have the discriminant:  $2|J_2| - |J_1| > 0$  and  $J_2 < 0$ . In the DFT calculation, the discriminant holds for 4 and 11



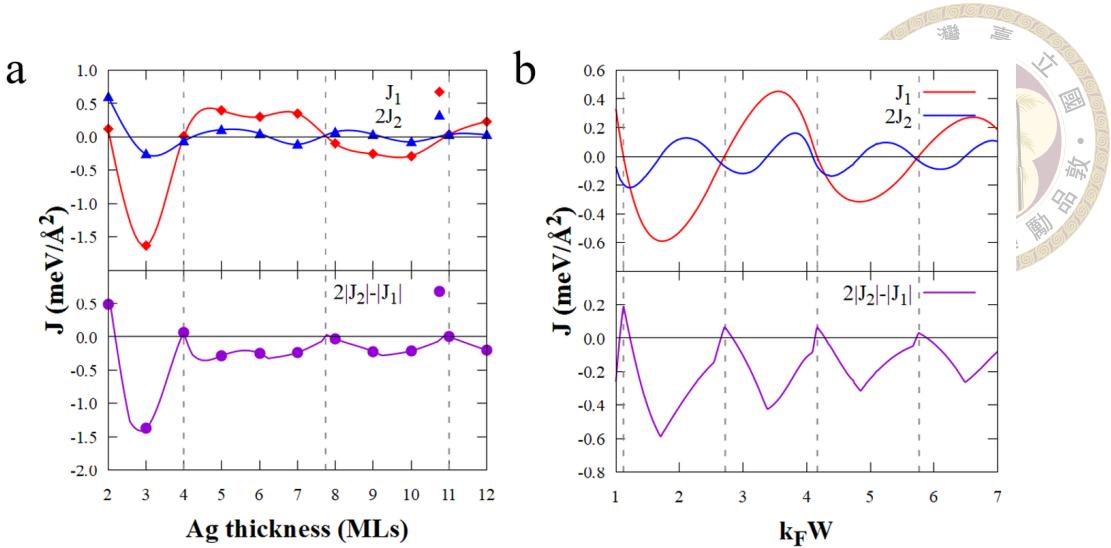


Figure 3.4: Interlayer exchange coupling between iron layers depends on the width of the intermediate silver layer. **a**, The figure above shows the bilinear and biquadratic coupling strengths  $J_1$  (red) and  $J_2$  (blue) calculated according to the first-principle. Straight lines connecting points are interpolated using cubic splines. At the bottom is the discriminant  $2|J_2| - |J_1|$  (purple). **b**, Analytical calculations of  $J_1$  (red),  $J_2$  (blue) and  $2|J_2| - |J_1|$  (purple). Where  $k_F$  is the Fermi wavevector in the silver layer and  $W$  is the width of the silver spacer layer.

MLs (see the bottom of Figure 3.4a). In fact, the energy of the three magnetic alignment in  $\text{Fe}_2\text{Ag}_{11}\text{Fe}_2$  is almost degenerate, and the energy difference is smaller than the numerical error. From the result and the discriminant, when the value of  $J_1$  is small, the noncollinear coupling is more likely to occur.

The band structure is essentially the same for both collinear and noncollinear calculations in parallel and antiparallel magnetic alignments. [39] More interesting for us is the electron in the transport direction. Thus, the band structures projected onto the  $p_z$  and  $d_{z^2}$  orbitals of the Ag atoms for  $W$  from 2 to 12 and three configurations; parallel (left), perpendicular (center) and antiparallel (right); are shown in Figs. 3.5-3.15. The color of the band represents the x component of the expected magnetization ( $m_x$ ). The antiparallel magnetic systems have  $C_2$  symmetry, so the bands are at least doubly degenerate. As the angle between the magnetic moments between the iron layers decreased to  $0^\circ$ , the symmetry broke and the bands split. The similar bands near the Fermi-energy rise with

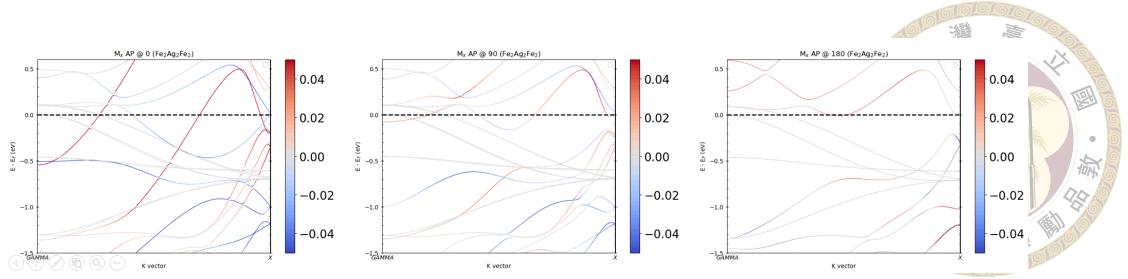


Figure 3.5: The projected band structures of the  $\text{Fe}_2\text{Ag}_2\text{Fe}_2$  trilayer for the three configurations, parallel (left), perpendicular (middle) and antiparallel (right). The color of the bands represents the x-component of the projected magnetization ( $m_x$ ).

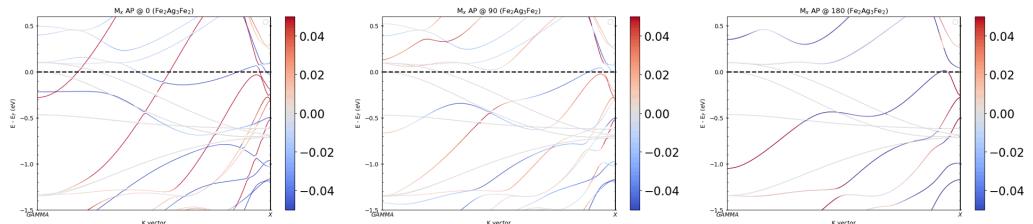


Figure 3.6: The projected band structures of the  $\text{Fe}_2\text{Ag}_3\text{Fe}_2$  trilayer for the three configurations, parallel (left), perpendicular (middle) and antiparallel (right).

increasing  $W$  as the length of the transport, i.e. the thickness of the Ag spacer channel, increases.

Taking the noncollinear magnetically oriented three-layer structure  $\text{Fe}_2/\text{Ag}_4/\text{Fe}_2$  as the research object, the energy band structure and spatial spin orientation of the silver quantum well are calculated to understand the characteristics of the electronic state in the non-collinear GMR system. The energy band structure of the Ag atoms  $p_z$  and  $d_{z^2}$  orbitals, relative to the transport direction, is shown in Figure 3.16a. Likewise, the color represents the x component of the projected magnetization ( $m_x$ ). The electron-like quantum well state is located at the  $\Gamma$  points of the two energy bands marked with circles. [37, 39, 46]

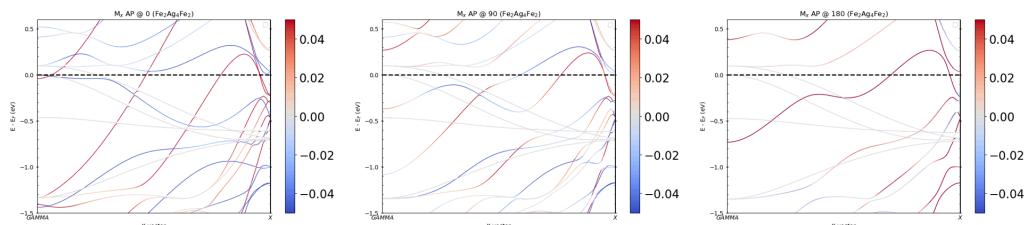


Figure 3.7: The projected band structures of the  $\text{Fe}_2\text{Ag}_4\text{Fe}_2$  trilayer for the three configurations, parallel (left), perpendicular (middle) and antiparallel (right).

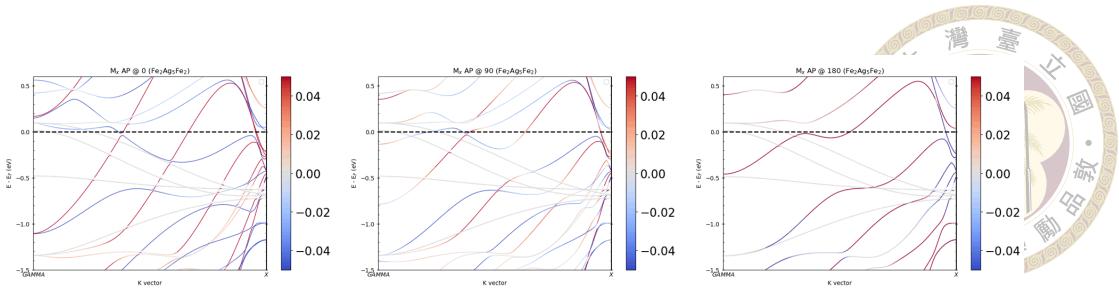


Figure 3.8: The projected band structures of the  $\text{Fe}_2\text{Ag}_5\text{Fe}_2$  trilayer for the three configurations, parallel (left), perpendicular (middle) and antiparallel (right).

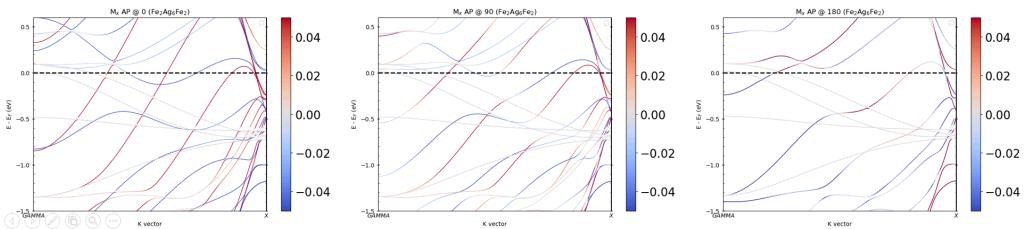


Figure 3.9: The projected band structures of the  $\text{Fe}_2\text{Ag}_6\text{Fe}_2$  trilayer for the three configurations, parallel (left), perpendicular (middle) and antiparallel (right).

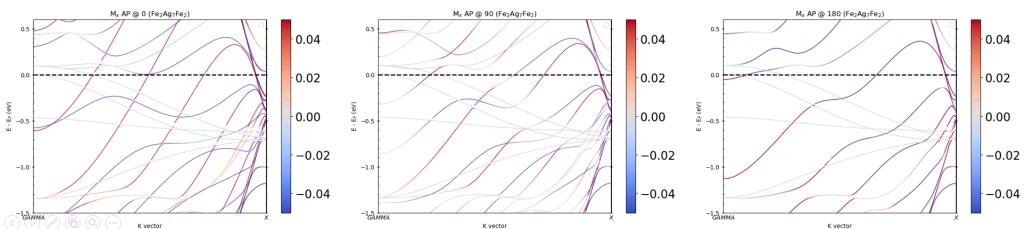


Figure 3.10: The projected band structures of the  $\text{Fe}_2\text{Ag}_7\text{Fe}_2$  trilayer for the three configurations, parallel (left), perpendicular (middle) and antiparallel (right).

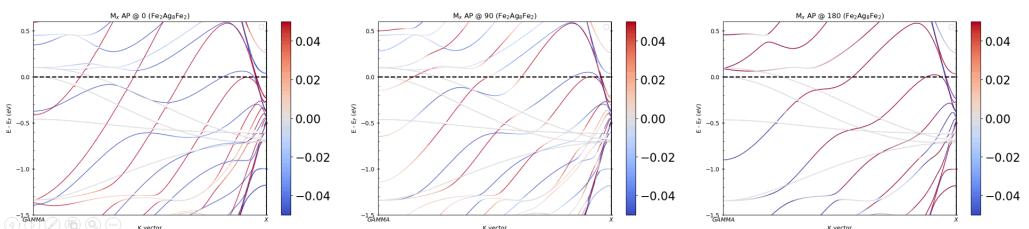


Figure 3.11: The projected band structures of the  $\text{Fe}_2\text{Ag}_8\text{Fe}_2$  trilayer for the three configurations, parallel (left), perpendicular (middle) and antiparallel (right).

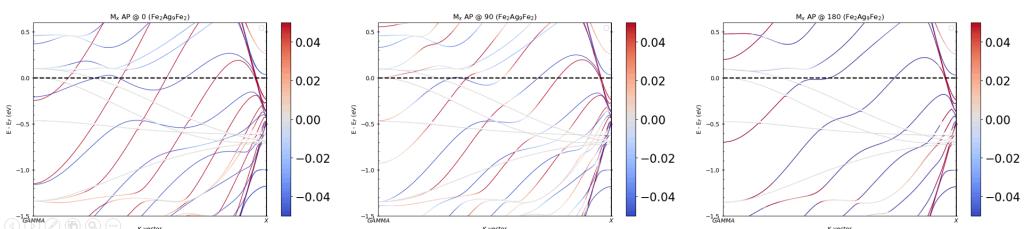


Figure 3.12: The projected band structures of the  $\text{Fe}_2\text{Ag}_9\text{Fe}_2$  trilayer for the three configurations, parallel (left), perpendicular (middle) and antiparallel (right).

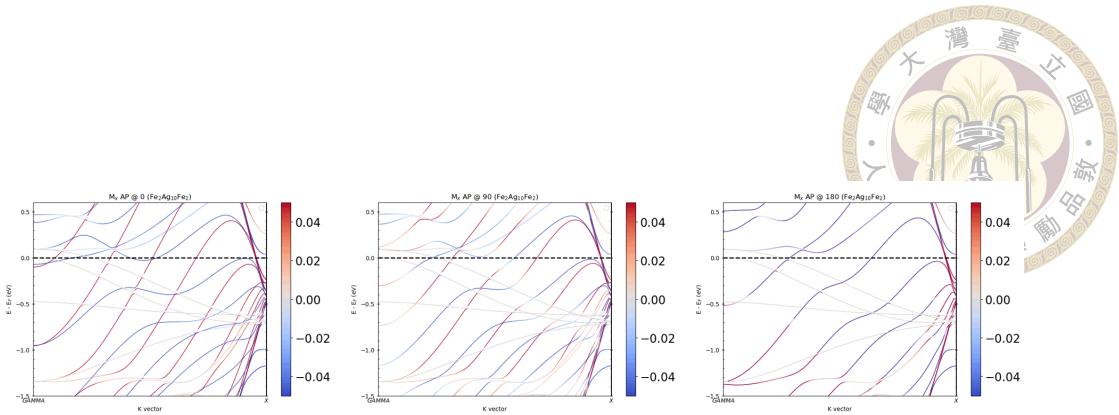


Figure 3.13: The projected band structures of the  $\text{Fe}_2\text{Ag}_{10}\text{Fe}_2$  trilayer for the three configurations, parallel (left), perpendicular (middle) and antiparallel (right).

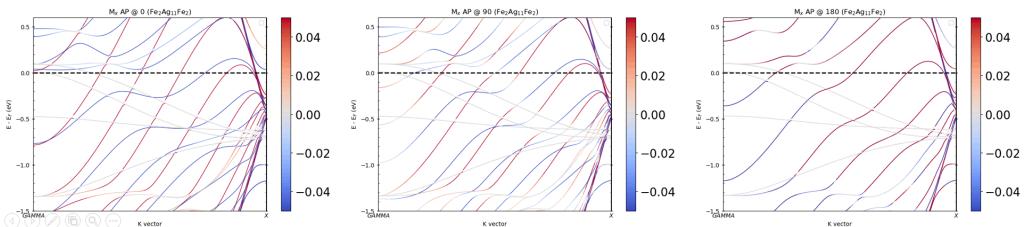


Figure 3.14: The projected band structures of the  $\text{Fe}_2\text{Ag}_{11}\text{Fe}_2$  trilayer for the three configurations, parallel (left), perpendicular (middle) and antiparallel (right).

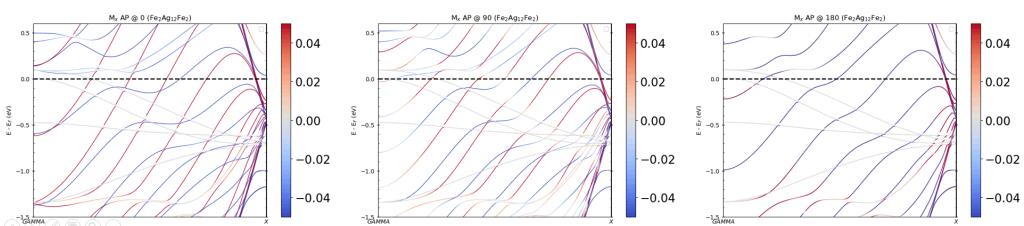
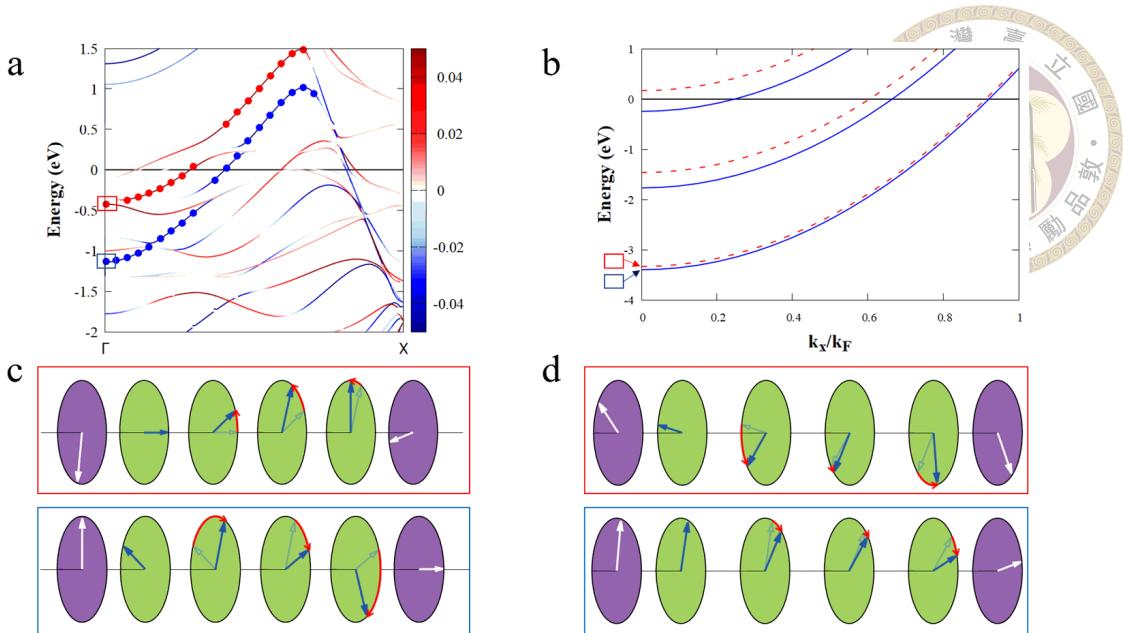
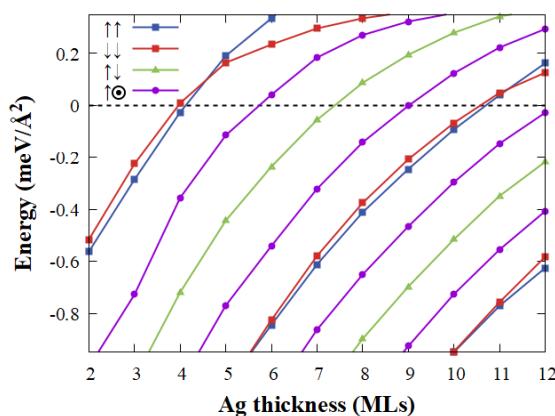


Figure 3.15: The projected band structures of the  $\text{Fe}_2\text{Ag}_{12}\text{Fe}_2$  trilayer for the three configurations, parallel (left), perpendicular (middle) and antiparallel (right).



**Figure 3.16: Perpendicular magnetic alignment system band structure and spin rotation diagram.** **a**, First-principles calculations of the projected band structure of the noncollinear coupling  $\text{Fe}_2/\text{Ag}_4/\text{Fe}_2$ . ( $p_z$  and  $d_{z^2}$ ) orbital of the silver atoms are projected. The color represents the  $x$  component of the expected magnetization ( $m_x$ ). The two QWS bands are represented by the solid circles. **b**, The analytically computed band structures. The QWS band with clockwise (counterclockwise) rotation of the spin is indicated by the blue (red) line. The spin rotation plots from **c**, ab initio, and **d**, analytically derived, respectively, in accordance with the squares in Figures **a** and **b**. Arrows indicate normalized rotation in the transverse direction, and the horizontal direction indicates the direction of transmission. The coloring of the full circles indicates the different layers: green for the spacing layer(silver) and violet for the magnetic layer(iron).

As shown in Figure 3.16c, the projected magnetization of these energy bands at the  $\Gamma$  point rotates along the transport axis of each atom. It is worth noting that these unique bands always seem to be paired with one another. These two rotations, clockwise  $90^\circ$  and counter-clockwise  $270^\circ$ , are the most striking observation to emerge from the data comparison, as they are exactly the two rotations that would result in a right  $90^\circ$  when less than a full rotation is made. This closely resembles the analytical findings on the right of the figures. (Figs. 3.16b and 3.16d). The results confirm that the spin orientation of Ag quantum well states in GMR systems with noncollinear magnetization is able to rotate in space, even in the absence of spin-orbital coupling. Figure 3.17 shows the minimum energy of the quantum well state energy band for three magnetic structures as a function



**Figure 3.17: Relationship between quantum well state energy and silver film width (**D**) in  $\text{Fe}_2/\text{Ag}_W/\text{Fe}_2$ .** For the parallel magnetized iron layers, the blue square represents the majority state of the quantum well and the red square represents the minority state of the quantum well. The green triangle corresponds to the quantum well state in the antiparallel arrangement of the iron layer.

of Ag thickness in order to relate the IECs to the quantum well resonance of the GMR system.

For the parallel magnetized iron layers, the blue square represents the majority state of the quantum well and the red square represents the minority state of the quantum well. The green triangle corresponds to the quantum well state in the antiparallel arrangement of the iron layer. Violet circles indicates the perpendicular arrangement of the iron layer. These lines connect energy points in the same quantum number. As the thickness of silver increases, the energy gap between pairs of the noncollinear alignment quantum states decreases.

When the minimum energy passes through the Fermi level (see the dotted line in Figure 3.17), the GMR system undergoes quantum resonance. For example, in a parallel magnetically aligned GMR system, if the Ag layer width is about 4 ml or 11 ml, quantum resonance occurs near the Fermi level (see blue and red squares in Figure 3.17). The quantum resonance period is about 7ML, which directly corresponds to the period of  $J_1$  in IEC (Figure 3.4a), which is consistent with previous work. [37, 39, 46] The IEC and

quantum resonance have the same period in the spacer layer, and the same is true in  $J_2$ . Quantum resonance occurs in GMR magnetic systems with spacer widths of about 6 MLs and 9 MLs in perpendicular arrangement (see purple circles in Figure 3.17). The quantum resonance period is about 3 MLs, corresponding to the period of  $J_2$  (see the top of Figure 3.4a). According to the direct correlation between Figure 3.4 and Figure 3.17, it is confirmed that by controlling the width of the GMR system, the perpendicular alignment of the GMR system can be essentially controlled.

### 3.6 Conclusion

In summary, we have calculated the bilinear and quadratic ICEs by using the first-principles method. This agrees with analytical calculations made by our co-worker. The results are also in agreement with previous experimental observations of the system. In addition, it has been found that the perpendicular magnetic configuration increases the degree of freedom in the choice of the magnetic structure as well as inducing the spatial spin orientation of the charge carriers. The results advance our understanding of nonlinear coupling and could aid the design of advanced spintronic devices.





# Chapter 4 Electronic properties of $\text{BiCu}_2/\text{Cu}(111)$

## 4.1 Introduction

Spintronics is the study of the intrinsic spin, associated magnetic moment, and fundamental electron charge of electrons in solids. Spintronics uses electron spin as an additional degree of freedom for charge states, making it fundamentally different from conventional electronics. This affects the efficiency of data storage and transmission.

The conductance of  $\text{BiCu}_2$  was measured by scanning tunneling spectroscopy (STS), which revealed that the conductance  $dI/dU$  is unusually large at 1.6 V.<sup>[47]</sup> We attempted to explain this phenomenon, but were unable to agree with the band structure calculated by the Korringa-Kohn-Rostoker method, which should be more accurate than the result from the Vienna Ab initio Simulation Package. Other groups have succeeded in deriving the accurate band structure and explaining the phenomenon, but we have learned a lot from this failure.

The spin topology in the  $\text{BiCu}_2$  surface is unconventional compared to other surface alloys with similar structure. Most of this type of surface alloy is agree with Rashba model,

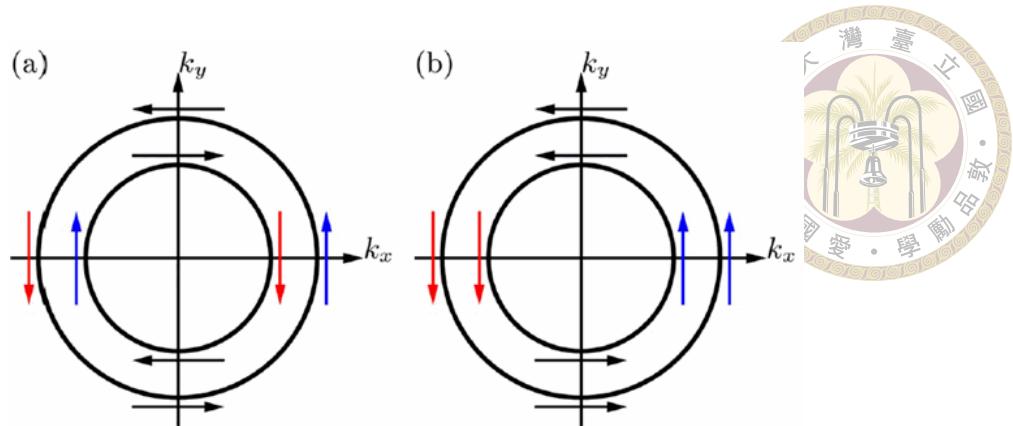


Figure 4.1: Spin-orbit separation 2DEG spin topology (schematic): (a) momentum distribution with traditional Rashba-type spin topology and (b) momentum distribution with unconventional spin topology. The "inner" and "outer" spin polarisations are indicated by arrows.

which explain the spin division and the antiparallel direction of the inner and outer spin, but both experiment (although it is not clear) and first principal calculation shows that the the direction of the inner and the outer spin are parallel in  $\text{BiCu}_2$  (see Figure 5.1). [6]

Band structures and density of states have been calculated for bulk  $\text{Cu}(111)$  from 2 layers up to 8 layers with  $\text{BiCu}_2$  on the surface. The unconventional spin topology bands we have do not agree with the result given by the Korringa-Kohn-Rostoker method, but the shape of the band structures were similar.

## 4.2 $\text{BiCu}_2$

The growth of Bi on the  $\text{Cu}(111)$  surface resulted in the formation of an ordered  $\text{BiCu}_2$  substitution surface alloy. [48] The structure for  $\text{BiCu}_2$  were accurately measured by Kaminski et al. [49] We have constructed the alloy by those structure parameters and then optimized by VASP (see Figure 4.2)

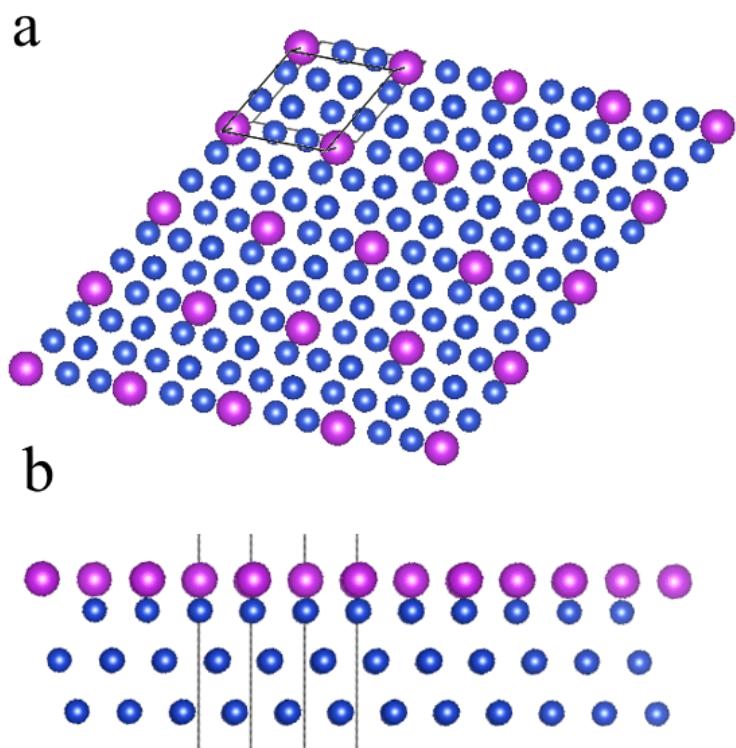


Figure 4.2: (a)Top view of BiCu<sub>2</sub> (b) Side view of BiCu<sub>2</sub>

### 4.3 Rashiba Effect

The Hamiltonian operator with spin-orbit is

$$H = \frac{1}{2}\hat{p}^2 + V(r) + \frac{1}{2c^2}\vec{\sigma} \cdot [\nabla V(r) \times \vec{p}] \quad (4.1)$$

$$= -\frac{1}{2}\nabla^2 + i\frac{\partial_z V(x)|_{z=0}}{2c^2}(\sigma_x \partial_y - \sigma_y \partial_x) \quad (4.2)$$

[50], which comes from



$$H_m = -\vec{\mu} \cdot \vec{B}$$

$$\vec{B} = -\frac{\vec{v} \times \vec{E}}{c^2} \quad (4.4)$$

$$\vec{v} = \frac{\vec{p}}{m} \quad (4.5)$$

$$\vec{E} = \nabla V \quad (4.6)$$

$$\vec{\mu} = -\frac{g_s \mu_B \vec{S}}{\hbar} = -\frac{g_s \mu_B \vec{\sigma}}{2} \quad (4.7)$$

This Hamiltonian is non-relativistic. We assume that the electrons in the alloy surface are a two-dimensional electron gas, i.e.  $V=V(z)$ , which is independent of the x, y component. Then we can simplify and extract the Rashiba term of the Hamiltonian

$$H_R = \alpha(\vec{\sigma} \times \vec{k}) \cdot \hat{z} \quad (4.8)$$

$$= \alpha(\sigma_x k_y - \sigma_y k_x) \quad (4.9)$$

$$= \alpha \begin{pmatrix} 0 & k_y + ik_x \\ k_y - ik_x & 0 \end{pmatrix} \quad (4.10)$$

Solving this Hamiltonian we get

$$E_{\pm} = \frac{\hbar^2 k^2}{2m^*} \pm \alpha |\vec{k}|, \vec{k} = (k_x, k_y) \quad (4.11)$$

Assume that the eigenfunctions consist of a plane-wave spinor,

$$\psi_k(r) = (\mu_k \chi^+ + \nu_k \chi^-) e^{i(k_x x + k_y y)} \quad (4.12)$$

The Pauli spinors  $\chi^\pm$  are quantized with respect to the z-axis, i.e,



$$\sigma_z \chi^\pm = \pm \chi^\pm \quad (4.13)$$

$$\sigma_x \chi^\pm = \pm \chi^\mp \quad (4.14)$$

$$\sigma_y \chi^\pm = \pm i \chi^\mp \quad (4.15)$$

Substituting  $\psi$  into Schrödinger's equation, we have

$$(k_y + ik_x)\nu_k^2 = (k_y - ik_x)\mu_k^2 \quad (4.16)$$

$$\mu_{k\pm} = \frac{1}{\sqrt{2}} \quad (4.17)$$

$$\nu_{k\pm} = \mp \frac{1}{\sqrt{2}} \sqrt{\frac{k_y - ik_x}{k_y + ik_x}}, \text{ for } k \neq 0 \quad (4.18)$$

The spin polarization is complete and normal to  $\mathbf{k}$ , so  $\mathbf{P}$  is written as follows

$$P_\pm(k) = \psi_{k\pm} \sigma \psi_{k\pm} \quad (4.19)$$

$$= \pm \frac{1}{|k|} \begin{pmatrix} k_y \\ k_x \\ 0 \end{pmatrix} \quad (4.20)$$

The above equation corresponds to the spin topology of Figure 4.1(a).

## 4.4 Result and Discussion

The DFT calculations were carried out with the Vienna Ab initio Simulation Package (VASP). [2, 43, 44] The pseudopotentials used were the Projector Augmented Wave (PAW) pseudopotentials [44] and the Perdew-Burke-Ernzerhof (PBE) generalized gra-

Table 4.1: The energy of the unit cell for different vacuum thicknesses.

Thickness (Å)	15	20	25	30
$E_0$ (eV/unit cell)	-32.0389	-32.0379	-32.0376	-32.0373

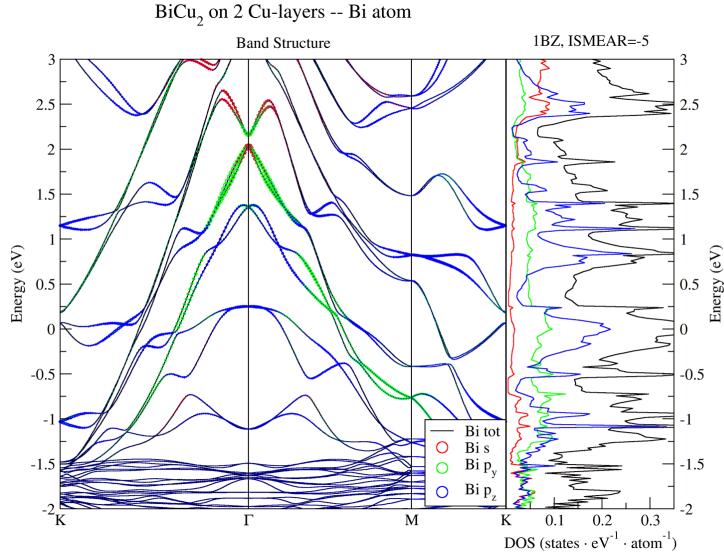


Figure 4.3: The projected band structures and density of states of  $\text{BiCu}_2$  on 2 layers of Cu(111) on Bi atoms. The size of the color circle indicates the weight of density of states of the s (red),  $p_y$  (green) and  $p_z$  (blue) orbitals.

dient approximation (GGA). Less than  $10^{-1}$  meV was used as the energy convergence criterion. The Brillouin zone was sampled using an  $11 \times 11 \times 1$  k-point mesh centred on gamma. A 450 eV energy cutoff was used for the plane-wave basis. The collinear calculation with an energy convergence criterion of 1 meV without spin-orbital coupling was used to obtain the structurally optimised unit cells. For the calculations of the band structure and the density of states, the spin-orbital coupling is included and the non-collinear calculation is carried out.

We have checked the convergence of the vacuum thickness. According to Table 4.1, 15 Å is thick enough for this system.

The projected band structures and density of states of  $\text{BiCu}_2$  on 2, 4, 6, and 8 layers of Cu(111) on Bi atoms are shown in Figs. 4.3-4.6. The size of the color circle indicates the weight of density of states of the s (red),  $p_y$  (green) and  $p_z$  (blue) orbitals.

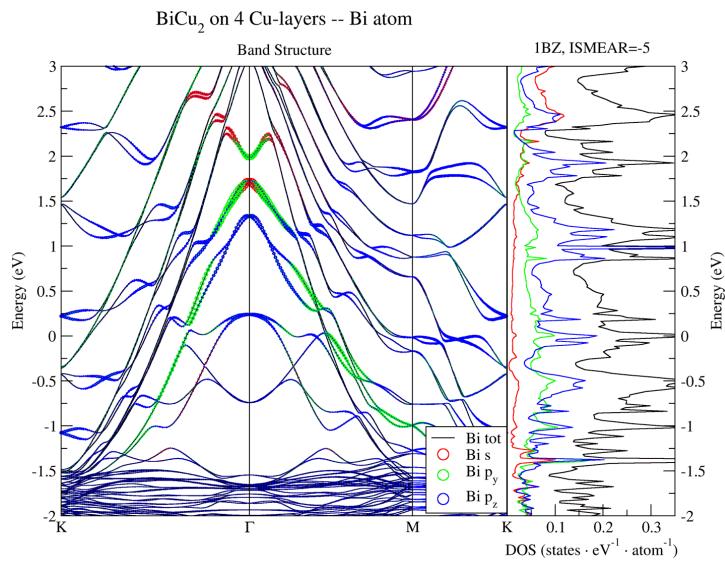


Figure 4.4: The projected band structures and density of states of BiCu<sub>2</sub> on 4 layers of Cu(111) on Bi atoms.

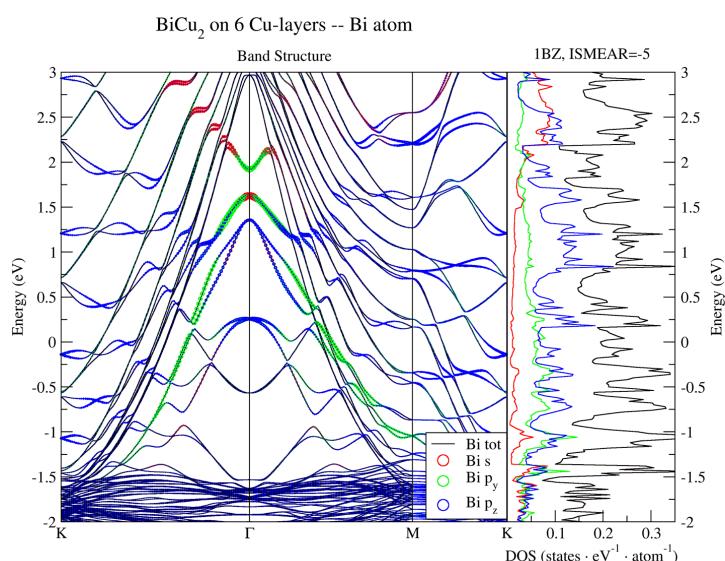


Figure 4.5: The projected band structures and density of states of BiCu<sub>2</sub> on 4 layers of Cu(111) on Bi atoms.

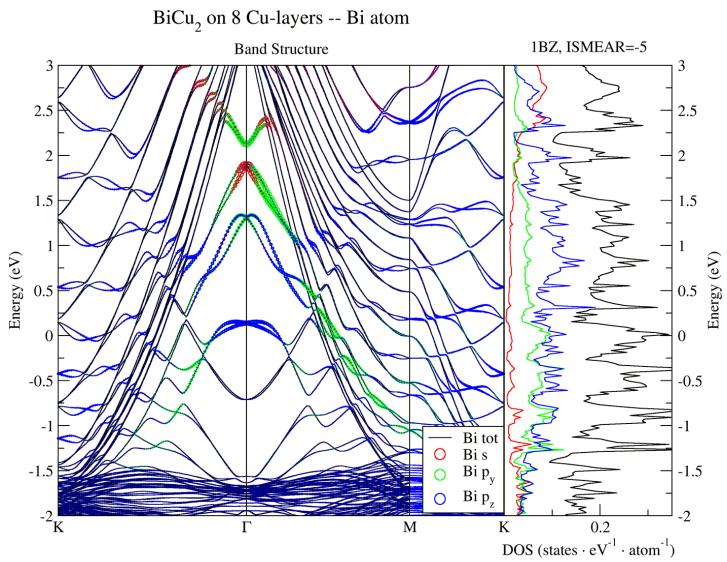


Figure 4.6: The projected band structures and density of states of  $\text{BiCu}_2$  on 4 layers of  $\text{Cu}(111)$  on Bi atoms.

The spin-resolved band structures of  $\text{BiCu}_2$  on 2, 4, 6, and 8 layers of  $\text{Cu}(111)$  for  $k_x$  are shown in Figs. 4.7-4.10. The color indicates the strength of y component of the magnetic moment.

The density of states does not increase significantly around 1.6 eV, but in the experiment the conductivity increases rapidly at 1.6 V. In general, there is an unconventional band around 1.7 eV. However, in the case of 6 Cu layers, the unconventional band almost becomes a conventional band. Another group, using the LDA exchange correlation functional and 9 layers of Cu, constrained the height of the surface Bi atom and obtained a better band structure and density of states. [47]

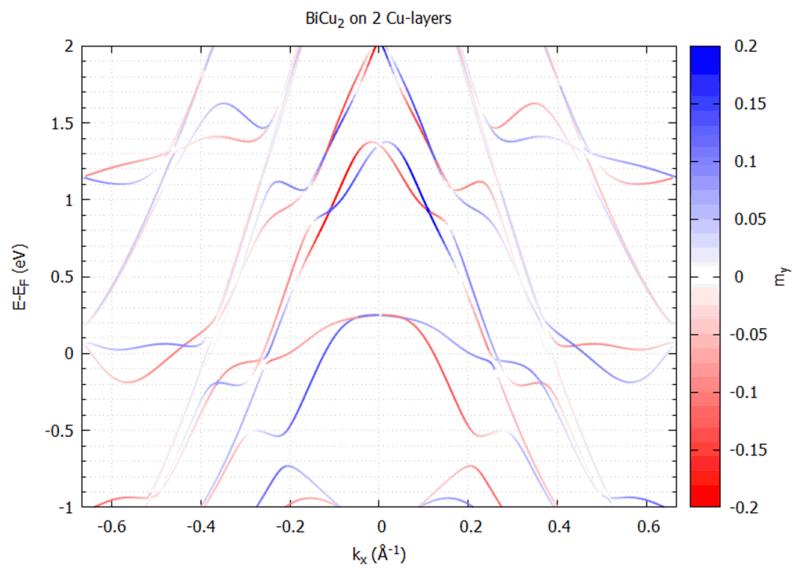
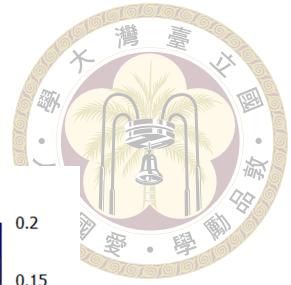


Figure 4.7: The spin-resolved band structures of BiCu<sub>2</sub> on 2 layers of Cu(111) for  $k_x$ . The color indicates the strength of  $y$  component of the magnetic moment.

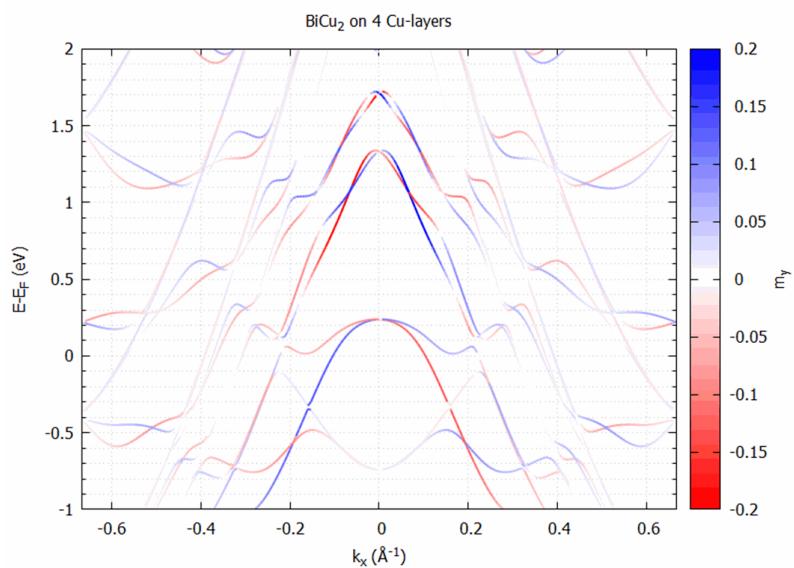


Figure 4.8: The spin-resolved band structures of BiCu<sub>2</sub> on 4 layers of Cu(111) for  $k_x$ .

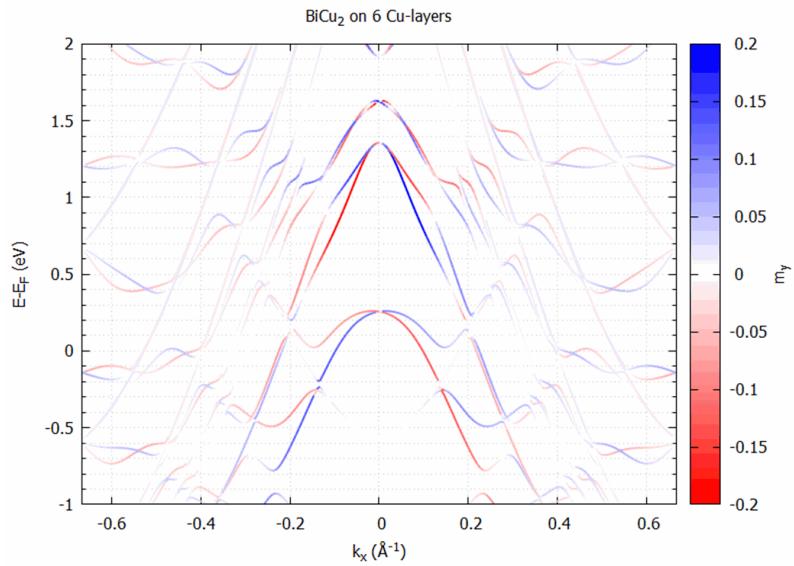


Figure 4.9: The spin-resolved band structures of BiCu<sub>2</sub> on 6 layers of Cu(111) for  $k_x$ .

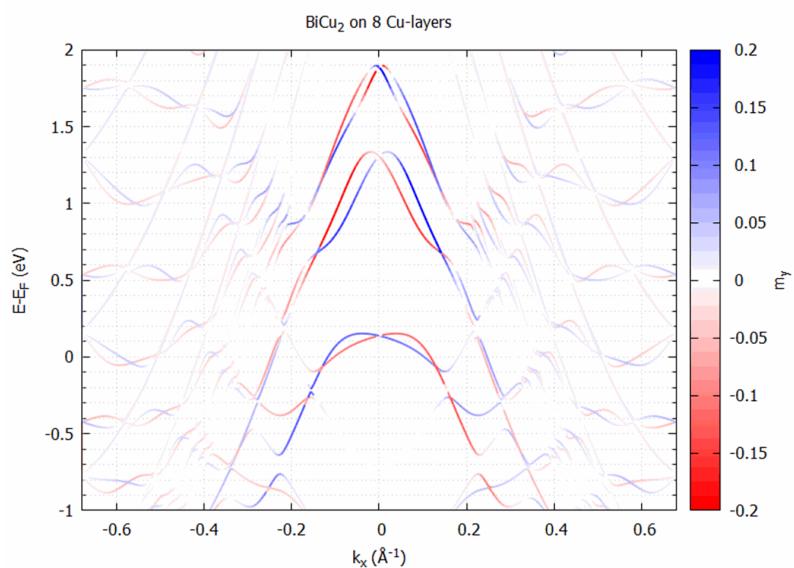


Figure 4.10: The spin-resolved band structures of BiCu<sub>2</sub> on 8 layers of Cu(111) for  $k_x$ .



# Chapter 5 Adsorption Energies of Chiral Peptides on an Au/Co Surface

## 5.1 Introduction

Magnetic force microscopy is a technique for detecting the magnetic moment on a surface. It is usually the magnetic dipole-dipole interaction between the tip of the probe and the surface of the material. The distance between the tip and the surface is relatively small because the strength of the interaction is proportional to  $r^{-3}$ . Amir Ziv et al. propose a new method using a chiral peptide to probe the spin moment of the surface. [7] The spin-exchange interaction between the chiral peptide and the surface is quite different for different spin orientations. The energy difference between different spin orientations. is about 152 meV.

We first constructed the unit cell including the tail of the peptide and the Au/Co surface. We then calculated the energy with different spin orientations, but the energy still did not agree with the experiment.



## 5.2 Chiral Induced Spin Selectivity Effect

The spin-exchange interaction between the chiral peptide and the surface could be the Chiral Induced Spin Selectivity (CISS) effect. As the chiral peptide approaches the surface, the magnetic moment from the surface acting on the chiral peptide redistributes the charge in the peptide. Both the electric and magnetic fields are induced by the redistribution of charge. The induced magnetic field is different for different chirals. This causes the energy to differ between the spin orientations. The Hamiltonian of the spin-orbital interaction can be described by

$$H_{SO} = \lambda \vec{\sigma} \cdot (\vec{p} \times \vec{E}_{\text{chiral}}) \quad (5.1)$$

## 5.3 Unit Cell of peptide on Au/Co

The chiral peptide is called alpha helix polyalanine (AHPA) in the reference. The chemical form of AHPA is HS-PEG-NH-AAAAKAAAAKAAAAKAAAAKAAAAKAAAAK- COOH. The whole peptide is too large to calculate, so we constructed only the tail, AAAAKAAAAKAAAAK- COOH. The positional information of the L-chiral peptide, AAAAKAAAAKAAAAK- COOH, is generated by "Avogadro", and optimized by VASP. The -COOH tail has been rotated to lower the oxygen atom, giving lower energy and a better expected interaction with the surface (see Figure 5.1 and Figure 5.2).

The surface in the experiment is coated with Au to prevent oxidation of Co. The lattice mismatch of Au/Co(111) is large for the small unit cell, but the large unit cell is too expensive to calculate. The strain energy is reduced by creating a Co transition layer.

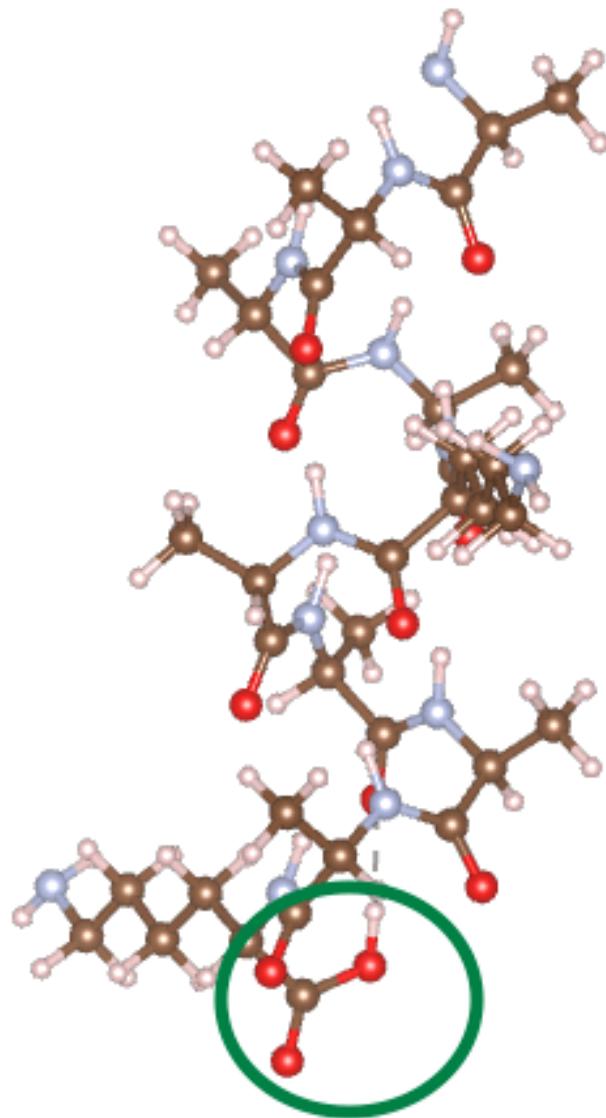


Figure 5.1: The L-AAAAKAAAAK-COOH molecule.

[51] The unit cell of the surface is constructed with one Au layer and two Co layers, and the one Co atom is removed and the Co atoms are rearranged in the first Co layer as a transition layer (see Figure 5.3).

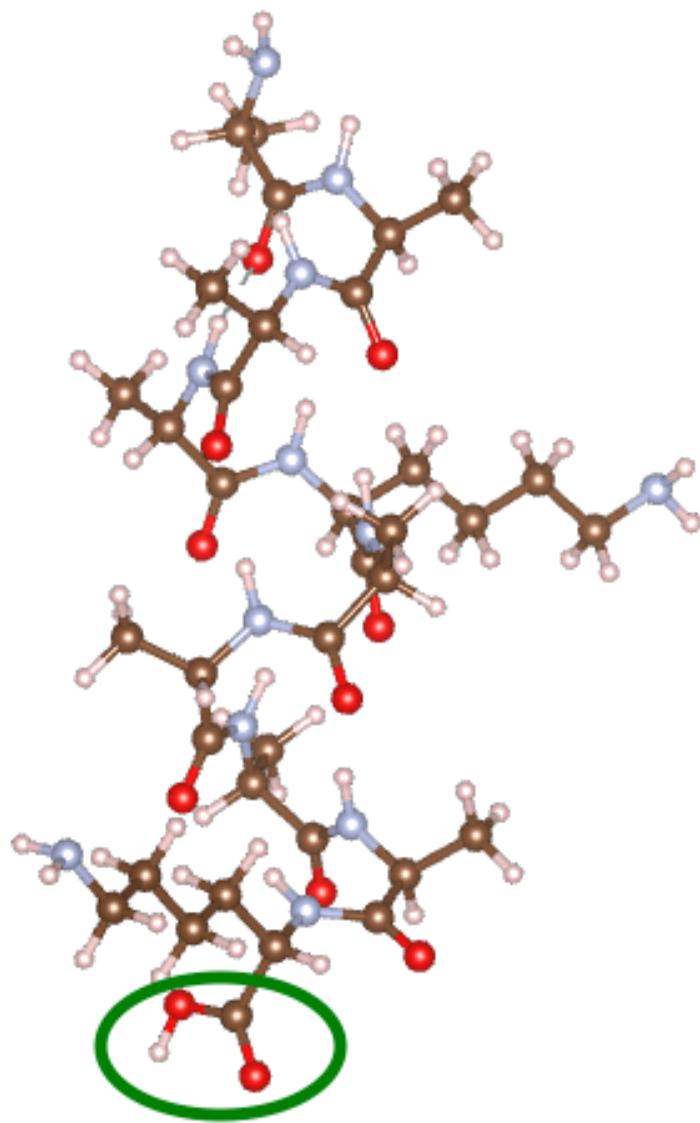


Figure 5.2: The L-AAAAKAAAAK-COOH molecule has undergone optimisation of the rotation of the -COOH tail.

## 5.4 Result and Discussion

The Vienna Ab initio Simulation Package (VASP) was used to perform the DFT calculations. [2, 43, 44], using PAW pseudopotentials and Perdew-Burke-Ernzerhof (PBE) GGA. [44] The criterion for the energy convergence has been set to be less than  $10^{-2}$  meV. The Brillouin zone was sampled using a  $1 \times 1 \times 1$  k-point gamma-centred mesh. The plane-wave basis energy threshold was set at 520 eV. The collinear calculation with

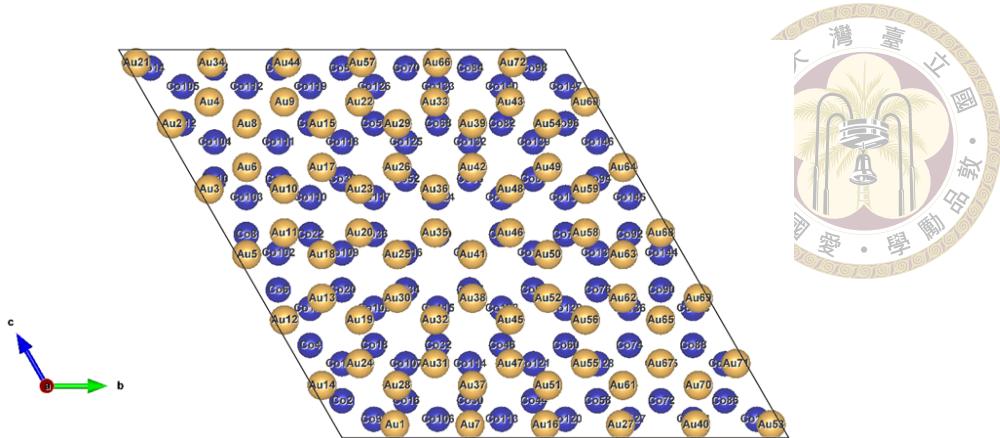


Figure 5.3: The unit cell of the peptide (L-AAAAKAAAAK-COOH) on the Au/Co surface.

Table 5.1: The total energy and magnetic moment with different spin.

Co spin orientation	Spin-up	Spin-down
Energy (eV)	-1918.762	-1918.762
Magnetic Moment (a.u.)	241.36	-241.36

the energy convergence criterion of  $10^{-1}$  meV without spin-orbital coupling was used to obtain the structurally optimised unit cells. Note that the Co atoms are fixed.

The total energy of L-AAAAKAAAAK-COOH on Au/Co(111) with different spin is given in Table 5.1. The energies are identical, so we include the spin-orbital coupling and recalculate (see Table 5.2).

The energy difference is still very small.

The height of the peptide plays an important role in this interaction, so we optimised the system with different heights of the peptide without spin-orbital coupling. The optimised height and total energy are listed in Table 5.3.

The energy difference at similar optimized height is still very small. We also tried to increase the length of the peptide, L-AAAAKAAAAKAAAAK-COOH, (see Table 5.4).

Table 5.2: The total energy with different spin including spin-orbital coupling.

Co spin orientation	Spin-up	Spin-down
Energy (eV)	-1940.282	-1940.273

Table 5.3: The optimized height and total energy with different spin

L-peptide	Spin-up						
Hight (Å)	3.942	3.689	3.5394	2.698	2.515	2.375	2.204
Energy (eV)	-1918.866	-1918.881	-1918.888	-1918.938	-1918.96	-1918.964	-1918.938
L-peptide	spin-dw						
Hight (Å)	3.942	3.693	3.538	2.698	2.507	2.344	2.123
Energy (eV)	-1918.867	-1918.877	-1918.889	-1918.934	-1918.954	-1918.956	-1918.908

Again, the results are not in agreement with the experiment.

Table 5.4: The total energy with different spin for L-AAAAKAKAKAKAKAK-COOH.

	Energy (eV)	Difference (eV)
Spin-up	-2286.327	
Spin-down	-2286.3251	0.0019
Spin-up with SOC	-2307.1045	
Spin-down with SOC	-2307.1055	-0.001

In order to observe the variation in electronic structure that is induced by the direction of the magnetic moment, the charge density difference for both cases is plotted.

$$\Delta\rho = \rho(\text{peptide on Au/Co}) - \rho(\text{peptide}) - \rho(\text{Au/Co}) \quad (5.2)$$

Figure 5.4 and Figure 5.5 illustrate the charge density difference of the peptide (L-AAAAKAKAKAK-COOH) on the Au/Co, with the direction of the spin of Co being up and down, respectively. The isosurface has been set to  $2 \times 10^{-4} e/\text{Bohr}^3$ . It is evident from the figures that the difference between the two cases is very similar to each other, as are the total energies (see Table 5.1).

The calculations demonstrate that the difference in electric properties between the spin of Co being up and down is not significant. This may be attributable to the inadequacy of the unit cell or the insufficient rigour of the energy convergence criteria.

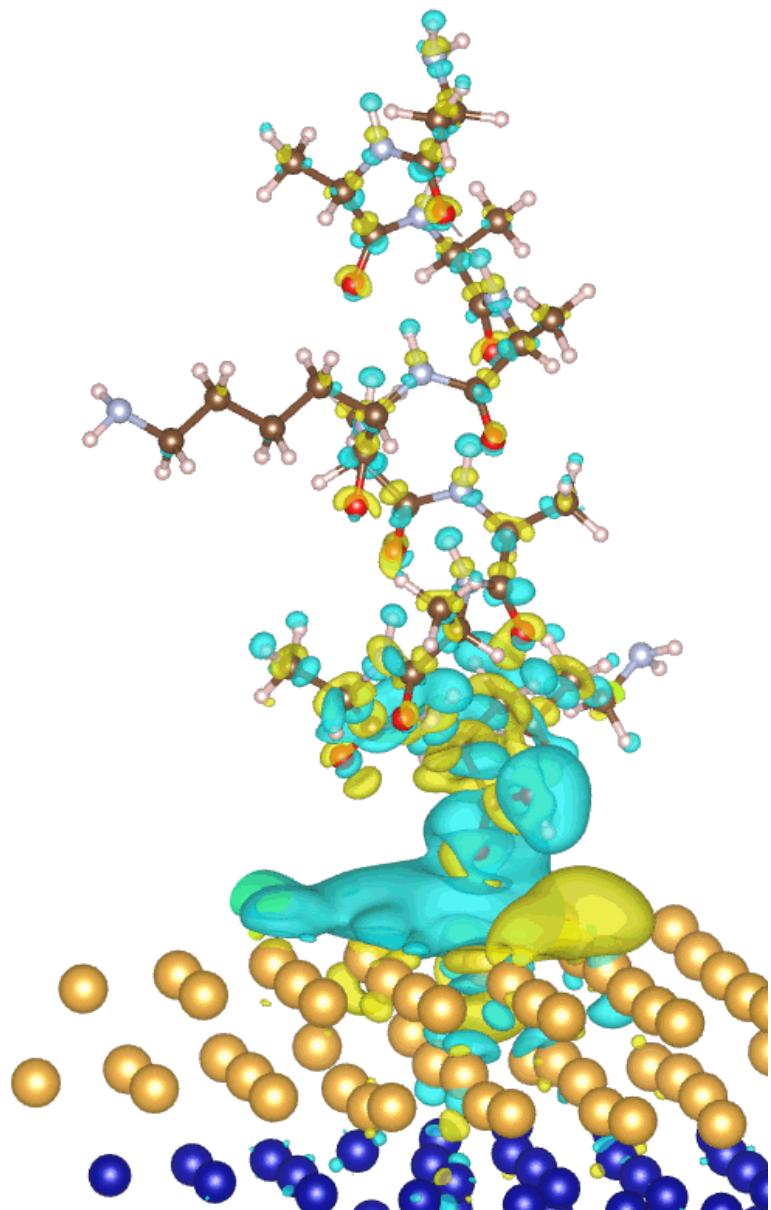


Figure 5.4: The charge density difference of the peptide (L-AAAAKAAAAK-COOH) on the Au/Co with Co spin-up.

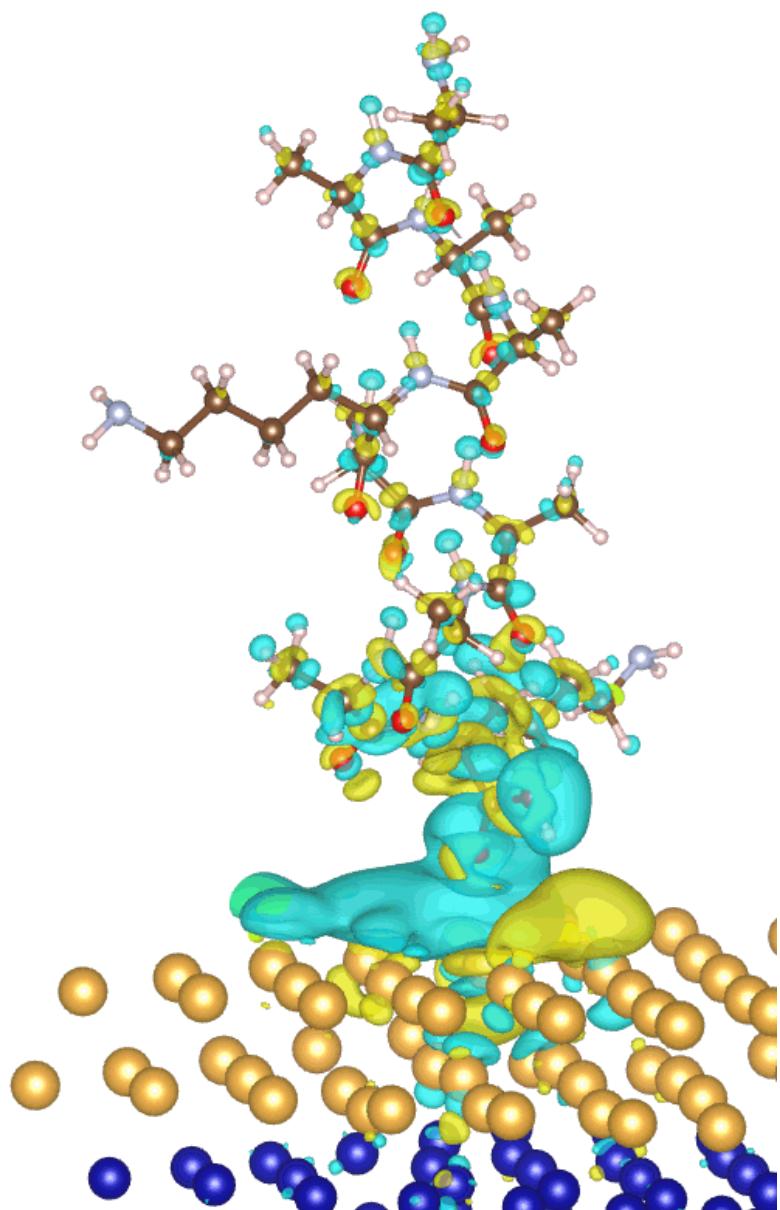
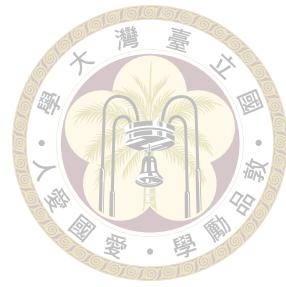


Figure 5.5: The charge density difference of the peptide (L-AAAAKAAAAK-COOH) on the Au/Co with Co spin-down.



## Chapter 6 Summary and Conclusion

In the Fe/Ag/Fe thin film system, the magnetic configurations of the ground state may be more than parallel and anti-parallel, with a perpendicular configuration also being a possibility. In the state with a perpendicular configuration, the rotation of the magnetic moment can also be in two ways: clockwise and anti-clockwise. These physical properties are of high potential application in spintronic devices.

The LDA exchange correlation energy functional was found to be superior in predicting the band structure of the  $\text{BiCu}_2$  alloy. However, GGA is typically more accurate in magnetic systems.

The following potential rationales may be postulated to explain the energy of the peptide on Au/Co: 1. The optimised distance between the peptide and the surface may have been erroneously calculated. 2. The energy convergence may have been inadequate to attain the ground state and correct shape of the peptide.

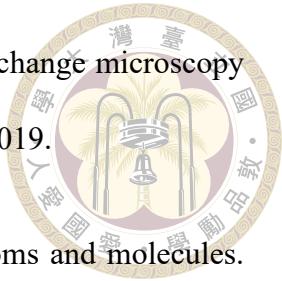




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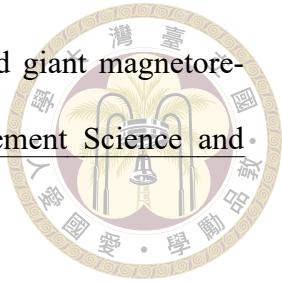
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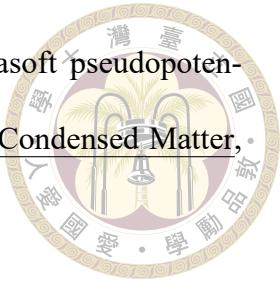
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